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**ORIGINAL ARTICLE** 

# Optical Absorption Study of Different Concentration of Nd<sup>3+</sup> ion in Zinc Lithium Bismuth Borate Glass

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## ABSTRACT

A new family of three optical glasses of (60-x)  $B_2O_3-10Bi_2O_3-20Li_2O-10ZnO-xNd_2O_3$  with x = 1, 1.5 and 2 were prepared by melt quenching technique. The amorphous nature of the prepared glass samples was confirmed by X-ray diffraction. The absorption spectra of three  $Nd^{3+}$  doped zinc lithium bismuth borate (ZLiBiB) glasses have been recorded at room temperature. Nine bands in the region 450-900 nm have been observed. Slater-Condon parameters  $F_k$  (k=2, 4 and 6), Lande's parameter  $\xi_{4f}$  and Racah parameters  $E^k$  (k=2, 4 and 6) have been computed. Using these parameters energies and intensities of these bands has been calculated. To study the nature of bonding in doped glasses nephelauxetic ratio ( $\beta'$ ) and bonding parameter ( $b^{1/2}$ ) have also been computed. The intensities of the f-f transitions in the absorption spectra have been analyzed by the application of the Judd-Ofelt theory. J-O parameters ( $\Omega_\lambda$ ) have been computed. The spectroscopic quality factor related with the rigidity of the glass system is aso discussed.

Keywords: Nd<sup>3+</sup>: ZLiBiB glasses, Energy interaction parameters, optical properties, Judd-Ofelt analysis

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## INTRODUCTION

Glasses containing heavy metal oxides (HMO) are of great interest due to their wide range of applications in the area of glass ceramics. Borate glasses containing unconventional  $Bi_2O_3$  as network modifiers possess high density, high refractive index, high optical basicity, extended far IR transmission and high non-linear optical susceptibility [1-4]. It has been made clear that  $B_2O_3$  could be used as a good network former (NWF) and other chemicals such as  $Bi_2O_3$ ,  $Li_2O$  and ZnO could be found as network modifiers (NWM) when these are added to the  $B_2O_3$  content. Bismuth oxide cannot be considered as a glass network former due to low field strength (0.53) of  $Bi^{3+}$  ion. However, in combination with  $B_2O_3$  glass former it is possible to obtain glasses in a relatively large compositional ranges. A survey of literature shows that there are many reports available on ternary bismuth borate glasses [5-8].

In rare earth doped borate glasses the central rare earth ion is surrounded by  $BO_3$  polymeric structure (boroxol ring) and  $BO_4$  units. When  $Bi_2O_3$  is added to the glass, a progressive conversion of  $BO_3$  units to  $BO_4$  units takes place [9, 10]. With the increase of  $BO_4$  units surrounding the central rare earth ion, the glass becomes more covalent, enhances the line strengths, effective fluorescence line widths decreases and the refractive index of the material increases. Covalency of the rare earth doped bismuth borate glass can be further increased [11-13] by co-doping the glass with Li<sub>2</sub>O and ZnO.

In the present work, the absorption spectra of Nd<sup>3+</sup>doped ZLiBiB glasses have been investigated. From the spectral data various energy interaction parameters like Slater-Condon parameters F<sub>k</sub> (k=2, 4 and 6), Lande parameter  $\xi_{4f}$  and Racah parameters E<sup>k</sup> (k=2, 4 and 6) have been computed. Nephelauxetic Ratio ( $\beta'$ ) and Bonding Parameters (b<sup>1/2</sup>) have also been computed from these parameters to study the nature of bonding in doped glasses. The intensity of the transition for the RE<sup>3+</sup> -ions has been calculated in term of Judd-Ofelt theory [14, 15]. This theory is known for a set of three intensity parameter  $\Omega_{\lambda}$  ( $\lambda = 2$ , 4 and 6) which are useful to the environment of trivalent rare earth ions. To understand the laser efficiency of these materials, the value of spectroscopy quality factor ( $\Omega_4/\Omega_6$ ) has been evaluated.

## EXPERIMENTAL

 $Nd^{3+-}$  zinc lithium bismuth borate glasses were synthesized by the conventional melt quenching method [16]. The starting chemicals used were reagent grade of  $H_3BO_3$ ,  $Bi_2O_3$ ,  $Li_2CO_3$  and ZnO,  $Nd_2O_3$  with more than 99.99% purity. The chemical compositions of the prepared glasses are as follows: (60-x)  $B_2O_3$ -10 $Bi_2O_3$ -

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 $20Li_2O-10ZnO-xNd_2O_3$  where x =1, 1.5 and 2 mol%. The raw materials were thoroughly mixed and ground in an agate pestle mortar in 15 g batches. The mixture was then melted in alumina crucibles in silicon carbide based an electrical furnace for 2h at 1050° C. The molten glass was then poured in a muffle furnace on to a steel mould and annealed at temperature of 350° C for 1 h to remove thermal strains and stresses. Every time fine powder of cerium oxide was used for polishing the samples. Optically transparent and bubble-free glasses were selected for optical studies.

1, 1.5 and 2 mol% of Nd<sup>3+</sup>: ZLiBiB glasses are labeled as ZLiBiB (1), ZLiBiB (1.5) and ZLiBiB (2) respectively. By Applying Archimede's principle, the glass densities were measured with Xylene as an immersion liquid on a single-pan electrical balance to the nearest 0.0001 g. The error in density measurement is estimated to be ±0.004 g cm<sup>-3</sup>. The refractive indices of these glasses have been measured at  $\lambda$  = 589.3 nm on an Abbe refractometer with an accuracy of  $\pm 0.001$ . The sample being glassy, it requires an adhesive coating on its surface, preferably 1-monobromonaphthalene as the contact layer between the sample and prism of the refractometer by using a sodium vapor lamp.

X-ray diffractogram of glass samples were studied with PANalytical X'pert Pro MPD diffractometer of  $CuK\alpha$ radiation (1.5406 Å) system. The absorption spectra of these glasses were recorded between wavelength ranges 450-900 nm with a Perkin-Elmer Lambda 750 UV/VIS/NIR Spectrophotometer at room temperature.

Table 1. Physical properties of Nd <sup>3+</sup> doped ZLiBiB glasses							
Physical properties	ZLiBiB (1)	ZLiBiB (1.5)	ZLiBiB (2)				
Refractive Index nd, at 589.3 nm	1.662	1.661	1.659				
Density, d (g/cm)	3.8602	3.8608	3.8610				
Thickness, t (cm)	0.304	0.309	0.308				
Average molecular weight $\overline{M}$ (g)	105.15	106.49	107.82				
Rare earth ions concentration N (×10 <sup>21</sup> ions/cms)	2.211	3.275	4.314				
Dielectric Constant (ε)	2.762	2.759	2.752				
Optical Dielectric Constant (pdt/dp)	1.762	1.759	1.752				
Molar Volume V <sub>m</sub> (gm/cm <sup>3</sup> )	27.240	27.582	27.925				
Mean atomic volume (gm/cm <sup>3</sup> /atom)	0.0633	0.0641	0.0649				
Reflection losses (R)	6.184	6.170	6.142				
Molar refractivity (R <sub>m</sub> )	10.080	10.194	10.297				
Polaron radius (r <sub>p</sub> )(A <sup>0</sup> )	3.093	2.714	2.475				
Interionic distance (r <sub>i</sub> )(A <sup>0</sup> )	7.676	6.734	6.143				
Electronic polarizability $\alpha_e$ (10-23 ions cm-3)	3.996	2.695	2.041				
Field strength F(10 <sup>16</sup> ×cm <sup>-2</sup> )	0.3178	0.4195	0.5028				



Fig.1. XRD Pattern of Nd<sup>3+</sup> : ZLiBiB glasses

## **Result and discussion**

The XRD patterns of all the glasses are shown in Fig.1, which confirmed its amorphous nature. The various physical properties of all the glass samples are presented in Table 1. An increase in the average molecular weight  $(\overline{M})$  influences significantly both the refractive index and density. However the electronic polarizability ( $\alpha_{e}$ ), polaron radius ( $r_{p}$ ), interionic distance ( $r_{i}$ ) slightly decreased with the increase of rare

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earth ions. The absorption spectra of Nd<sup>3+</sup> doped ZLiBiB glass specimens have been presented in Fig. 2 in terms of relative absorption log(I<sub>0</sub>/I) versus wavelength (nm), where I and I<sub>0</sub> are intensities of the radiation transmitted through doped specimens and undoped specimens of equal thickness. Nine absorption bands have been observed from the ground state  ${}^{4I_{9/2}}$  to excited states  ${}^{4F_{3/2}}$ ,  ${}^{4F_{5/2}}$ ,  ${}^{4F_{7/2}}$ ,  ${}^{4F_{9/2}}$ ,  ${}^{2H_{11/2}}$ ,  ${}^{4G_{5/2}}$ ,  ${}^{4G_{7/2}}$ ,  ${}^{4G_{9/2}}$ , and  ${}^{2G_{9/2}}$  for Nd<sup>3+</sup> doped ZLiBiB glasses.



Fig.2. Vis-NIR absorption spectra of Nd3+:ZLiBiB glasses

The absorption spectra of lanthanide ions lie in the visible and near infrared regions and correspond to transitions from the ground state to various excited states of  $4f^N$  configuration. The various energy interaction parameters viz. Slater-Condon parameters  $F_k$  (k=2, 4 and 6), Lande parameter  $\xi_{4f}$  and Racah parameters  $E^k$  (k=2, 4 and 6) have been computed using partial regression method and formulae described elsewhere [17]. The computed values of these parameters for Nd<sup>3+</sup> doped ZLiBiB glass specimens along with those for the free ion for comparison have been presented in Table 3.

Table 2- Experimental energy ( $E_{exp}$ ) and calculated energy ( $E_{cal}$ ) with their difference ( $\Delta E$ ) for various absorption bands of									
Nd <sup>3+</sup> do	doped ZLiBiB glass specimens of differer ZLiBiB (1)			ent doping co Z	t doping concentrations ZLiBiB (1.5)			ZLiBiB (2)	
Absorption level s	Em (cm <sup>-1</sup> )	Ec (cm <sup>-1</sup> )	ΔE (cm <sup>-1</sup> )	Em (cm <sup>-1</sup> )	Ec (cm <sup>-1</sup> )	ΔE (cm <sup>-1</sup> )	Em (cm <sup>-1</sup> )	Ec (cm <sup>-1</sup> )	ΔE (cm <sup>-1</sup> )
<sup>4</sup> F <sub>3/2</sub>	11442	11402	40	11435	11394	41	11448	11410	38
<sup>4</sup> F <sub>5/2</sub>	12469	12469	0	12461	12464	-3	12469	12476	-7
<sup>4</sup> F <sub>7/2</sub>	13405	13308	97	13396	13302	94	13405	13313	92
<sup>4</sup> F <sub>9/2</sub>	14706	14772	-66	14706	14770	-64	14717	14778	-61
$^{2}H_{11/2}$	15974	15974	0	15974	15975	-1	15987	15987	0
${}^{4}G_{5/2}$	17167	17176	-9	17167	17167	0	17197	17191	6
<sup>4</sup> G <sub>7/2</sub>	19048	19155	-107	19048	19158	-110	19066	19176	-110
${}^{4}G_{9/2}$	19531	19558	-27	19531	19561	-30	19550	19575	-25
${}^{2}G_{9/2}$	21209	21158	51	21222	21169	53	21231	21180	51
rms deviation (σ)		±58.090			±58.205			±56.844	

The values of F<sub>2</sub> parameters exhibit a reduction of 1.22% in glass ZLiBiB (1), 1.703% in glass ZLiBiB (1.5) and 1.455% in glass ZLiBiB (2) from the corresponding free ion value.  $F_4/F_2 \sim (0.154-0.152)$  and  $F_6/F_2 \sim (0.0157-0.0156)$  are nearly same as reported in other glasses [18]. The values of E<sup>1</sup>, E<sup>2</sup> and E<sup>3</sup> parameters have been given in Table 3. The ratio of E<sup>1</sup>/E<sup>3</sup> ~ (10.16-10.13) and E<sup>2</sup>/E<sup>3</sup> ~ (0.0484-0.0478) are found to remain almost constant over the entire range of Nd<sup>3+</sup> doping concentrations and are in good agreement with the corresponding hydrogenic ratios.

The values of nephelauxetic ratio  $\beta'$  and bonding parameter  $b^{1/2}$  for Nd<sup>3+</sup> ion in ZLiBiB glasses indicate covalent nature of bonding between the rare earth ion and the surrounding glass matrix. Small values of r.m.s. deviation ' $\sigma$ ' between experimental energy ( $E_{exp}$ ) and calculated energy ( $E_{cal}$ ) of absorption levels in

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Table 3- Calculated values of Slater-Condon, Lande's, Racah, Nephelauxetic ratio and bonding parameters of Nd <sup>3+</sup> doped ZLiBiB glass specimens of different doping concentrations						
Parameters	Free ion	ZLiBiB (1)	ZLiBiB (1.5)	ZLiBiB (2)		
F <sub>2</sub> (cm <sup>-1</sup> )	331.16	327.12	325.52	326.34		
F4 (cm <sup>-1</sup> )	50.71	49.95	50.25	50.29		
F <sub>6</sub> (cm <sup>-1</sup> )	5.154	5.147	5.087	5.109		
ξ <sub>4f</sub> (cm <sup>-1</sup> )	884.0	893.46	893.90	892.08		
$E^{1}$ (cm <sup>-1</sup> )	5024.0	4971.33	4953.20	4965.75		
$E^{2}$ (cm <sup>-1</sup> )	23.90	23.701	23.37	23.468		
E <sup>3</sup> (cm <sup>-1</sup> )	497.0	488.96	488.73	489.52		
$F_4/F_2$	0.1531	0.1527	0.1543	0.1541		
$F_6/F_2$	0.0155	0.015735	0.015627	0.015656		
$E^{1}/E^{3}$	10.1086	10.1671	10.1347	10.1441		
$E^2/E^3$	0.0481	0.048471	0.047829	0.047941		
β'	-	0.9878	0.9829	0.9854		
b <sup>1/2</sup>	-	0.0781	0.0922	0.0852		

neodymium ion doped LBB glass specimens justify the suitability of the use of Taylor series expansion method [19, 20].

From the observed absorption spectra, the intensities of the observed bands have been calculated in terms of oscillator strengths,  $f_{exp}$ , and line strengths,  $s_m$ , spectral region wise and have been collected in Table 3. The Judd-Ofelt [14, 15] intensity parameters  $\Omega_{\lambda}$  ( $\lambda = 2, 4$  and 6) for all the glasses have been computed using line strengths and matrix element values by partial regression method [21] and have been presented in Table 5. The intensities of all the nine observed bands have been included in this calculation. Computation of  $\Omega_{\lambda}$  parameters is very important since they have been further used in the calculation of different radiative properties. The calculated line strengths agree well with the experimental values. For Nd<sup>3+</sup> doped ZLiBiB glasses,  $\Omega_{\lambda}$  values vary as  $\Omega_2 > \Omega_6 > \Omega_4$ . The  $\Omega_2$  parameter is affected by the covalency, the  $\Omega_6$  parameter is related to the rigidity of the glass hosts, and  $\Omega_4$  parameters are mainly due to the hypersensitive transition  ${}^{4I_{9/2} \rightarrow 4}G_{5/2}$ . The bonding environment surrounding the rare earth ion has been discussed on the basis of the value of  $\Omega_{\lambda}$  parameters. The success of Judd-Ofelt theory has been shown by low value of goodness of fit between the measured (S<sub>m</sub>) and calculated (S<sub>c</sub>) line strengths.

Table 4- Experimental oscillator strength (fexp), measured (Sm) and calculated (Sc) absorption line strength for Nd <sup>3+</sup> doped									
ZLiBiB glass specimens of different doping concentrations									
	ZLiBiB (1) ZLiBiB (1.5)		1.5)	ZLiBiB (2)					
Absorption Levels	f <sub>exp</sub> (10 <sup>-6</sup> )	S <sub>m</sub> (10 <sup>-20</sup> ) cm <sup>2</sup>	S <sub>c</sub> (10- <sup>20</sup> )	f <sub>exp</sub> (10 <sup>-6</sup> )	S <sub>m</sub> (10 <sup>-20</sup> ) cm <sup>2</sup>	S <sub>c</sub> (10- <sup>20</sup> )	f <sub>exp</sub> (10 <sup>-6</sup> )	S <sub>m</sub> (10 <sup>-20</sup> ) cm <sup>2</sup>	S <sub>c</sub> (10 <sup>-</sup> <sup>20</sup> )
			CIII2			CIIIs			CIII2
${}^{4}F_{3/2}$	2.224	1.182	1.047	2.153	1.146	0.997	2.015	1.073	0.947
<sup>4</sup> F <sub>5/2</sub>	7.206	3.514	3.698	6.424	3.137	3.354	6.283	3.071	3.293
<sup>4</sup> F <sub>7/2</sub>	8.423	3.820	3.813	7.336	3.332	3.393	7.278	3.309	3.378
<sup>4</sup> F <sub>9/2</sub>	0.584	0.241	0.267	0.546	0.226	0.239	0.481	0.199	0.237
<sup>2</sup> H <sub>11/2</sub>	0.155	0.059	0.067	0.124	0.047	0.060	0.095	0.036	0.060
<sup>4</sup> G <sub>5/2</sub>	23.180	8.210	8.279	21.589	7.652	7.768	20.384	7.224	7.348
<sup>4</sup> G <sub>7/2</sub>	3.004	0.959	1.141	2.204	0.704	1.073	2.083	0.666	1.022
<sup>4</sup> G <sub>9/2</sub>	1.018	0.317	0.546	0.623	0.194	0.505	0.612	0.191	0.488
${}^{2}G_{9/2}$	0.873	0.386	-0.136	0.473	0.136	0.365	0.392	0.113	0.348
Goodness of fit		0.02694			0.06193			0.05965	

The spectroscopic quality factor  $(\Omega_4/\Omega_6)$  related with the rigidity of the glass system has been found to lie between 0.58 and 0.64 in the present glasses. This shows that these glasses are fairly rigid as compared to CdBiB [24] and zinc fluoride borophosphate [25] glasses.

Table 5- Judd-Ofelt intensity parameters, $\Omega_{\lambda}$ for Nd <sup>3+</sup> doped ZLIBIB glass specimens of different doping								
concentration and compare with other similar glasses.								
Glass Specimen	$\Omega_2(pm^2)$	$\Omega_4(\text{pm}^2)$	$\Omega_6(\text{pm}^2)$	$\Omega_4/\Omega_6$	Ref.			
ZLiBiB (1)	5.923	3.239	5.551	0.583	р			
ZLiBiB (1.5)	5.508	3.170	4.919	0.644	Р			
ZLiBiB (2)	5.212	2.952	4.911	0.601	р			
CdBiB	4.893	2.976	5.789	0.514	24			
ZFBP	1.46	0.38	0.75	0.50	25			

## **CONCLUSIONS**

In summary, it is concluded that we have successfully developed transparent, moisture resistant and more stable (60-x)  $B_2O_3$ -10 $Bi_2O_3$ -20 $Li_2O$ -10ZnO-xNd<sub>2</sub>O<sub>3</sub> glasses where x =1, 1.5 and 2 mol% for their optical characterization. The calculated energies of the transitions using least square fit method are exactly coinciding with the experimentally obtained energies of the transitions, reflecting the good accuracy of the experimentation. The hypersensitive transition  ${}^{4}I_{9/2} \rightarrow {}^{4}G_{5/2}$  exhibits high intensity as a function of Judd-Ofelt intensity ( $\Omega_2$ ) parameter in the case of 1 mol% Nd<sup>3+</sup> glass system. The spectroscopic quality factor related with the rigidity of the glass system has been found to lie between 0.58 and 0.64 in the present glasses, which is fairly large as compared to other glasses. The large values of spectroscopic quality factor  $(\Omega_4/\Omega_6)$  make them suitable for optical devices.

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## **Ciation of this article**

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